

FERRITE MATERIALS, METHODS OF PREPARING THE SAME, AND PRODUCTS FORMED THEREFROM

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Technical Field

The present invention relates to ferrite materials, and, more particularly,
to manganese-zinc ferrite materials, methods of forming the same, and
10 products made therefrom.

Background

Ferrite materials, such as manganese-zinc ferrite compounds, have been
widely used as magnetic core materials for transformers in power supply
15 systems, as well as for household electric appliances, communication and
telecommunication equipment, computer and peripheral equipment, electronics
finished products, electronic components, and other products that employ high
frequency electronic circuitry. Ferrite materials have been found to exhibit
properties such as, high permeability, high saturation magnetic flux density,
20 high temperature stability, and low power losses that make these materials
suitable for high frequency applications. For example, in transformer
applications for power supply systems, sintered ferrite materials provide
relatively low power losses and high temperature stability when used at
relatively high switching frequencies. Typically, with switching frequencies
25 ranging from 100 kHz to 500 kHz, power losses are measured to be about 300
milliwatts per cubic centimeter (mW/cm^3) or greater, and Curie temperatures
range from 230°C to 240°C. As used herein, Curie temperature refers to the
critical temperature at which ferrite materials substantially lose their magnetic
characteristics. It is the combined benefits of relatively low power losses and

high temperature stability, for example, that make ferrite compositions particularly well suited for various and wide ranging high frequency electrical applications.

Due, in part, to the increased demand for employing high frequency
5 electronic circuitry into a wide range of components and equipment, efforts
have been made to advance the ways in which power supplies can be improved
and/or miniaturized for integration into these applications. This demand, at
times, has been tempered by the premium that is placed on the available space
inside these components. Typically, these efforts are directed to improving the
10 ability of the power supply to perform at high temperatures and high frequencies
with low core power losses, so that the size can be reduced without sacrificing
performance or operation. Thus, much of the attention devoted to the
miniaturization process is related to improvements to the material properties of
the ferrite materials, and is based on the equation $P \sim f B A$, wherein throughput
15 power (P) is proportional to operating frequency (f), magnetic flux density (B),
and magnetic cross section (A). Accordingly, increases in operating frequency
and/or magnetic flux density allow for reductions in magnetic cross section
without sacrificing throughput power. However, one disadvantage of operating
power supplies at higher frequencies relates to a corresponding increase in
20 core power losses that often limit the throughput power and result in an
overheating of the core. Thus, improving the material properties of ferrite
materials should also take into account effects on power losses.

Numerous attempts have been made to improve the chemical
formulations of ferrite compositions, or the process conditions in which these
25 compositions are sintered, in order to improve their material properties and
allow these materials to operate at higher temperatures and higher frequencies
with limited power losses. Some of these attempts are disclosed in U.S. Patent
Nos. 3,415,751, 3,481,876, 3,652,416, 3,769,219, 5,143,638, 5,368,763,
5,518,642, and 5,846,448. These patents disclose the use of various amounts

and combinations of Fe_2O_3 , MnO , and ZnO as major components, and one or more of Nb_2O_5 , CaO , SiO_2 , V_2O_5 , ZrO_2 , Al_2O_3 , SnO_2 , CuO , Co_3O_4 , TiO_2 , Co_2O_3 , Li_2O , Sb_2O_3 , Ta_2O_5 , for example, as minor components, at various processing conditions, such as sintering temperatures and pressures, that are said to

5 provide improved properties to the ferrite material. One such objective of these attempts is to enhance the resistivity of the ferrite material by improving the grain boundary resistivity and the resistivity of the ferrite grains themselves. For example, along with the major components, prior art compositions for high frequency applications have included relatively large amounts of Co_3O_4 , SnO_2 ,

10 TiO_2 , CaO , and the like, or combinations thereof, as minor components, in order to achieve certain material properties and characteristics.

In particular, a common approach to reduce powder losses is to increase the resistivity of the ferrite material in order to reduce eddy current losses at high frequencies. The various auxiliary additives, discussed above, in

15 combination with Fe_2O_3 , MnO , and ZnO have been investigated to achieve this objective. For example, one known composition that is used that is said to improve high frequency losses at frequencies up to 5 MHz includes 55-59 mol % Fe_2O_3 , 35-42 mol % MnO , and 6 mol % or less of ZnO , with additions of 0.050-0.300 wt% CaO , 0.005-0.050 wt% SiO_2 , and 0.010-0.200 wt% of one or

20 more of the following: ZrO_2 , Ta_2O_5 , MoO_3 , In_2O_3 , Sb_2O_3 , and Bi_2O_3 . Grain size of 2-5 μm is preferred in the final sintered body. Compositions outside of these ranges were are said to have higher power losses and lower minimum power loss temperatures. Representative examples of these compositions are shown in Tables 1 and 2, as comparative examples 20 - 22.

25 However, it has been found that the prior art materials are difficult to sinter and achieve consistent material properties because of their sensitivity to firing conditions. Thus, there is a continued need to provide ferrite compositions having improved and consistent material properties, such as high temperature stability and low power loss when used at relatively high

frequencies, that allow for improvements in the manufacture and performance of high frequency related compounds that incorporate these materials.

Summary

5 In one embodiment, the present invention provides a ferrite material including, as main components, an iron oxide ranging from 55.5 to 58.0 mole percent calculated as Fe_2O_3 , an amount of manganese oxide ranging from 38.0 to 41.0 mole percent calculated as MnO , and an amount of zinc oxide ranging from 3.3 to 4.7 mole percent calculated as ZnO . The present invention also
10 includes, as minor components, an amount of calcium oxide ranging from 0.030 to 0.100 weight percent calculated as CaO , an amount of silicon oxide ranging from 0.015 to 0.040 weight percent calculated as SiO_2 , and an amount of niobium oxide ranging from 0.010 to 0.030 weight percent calculated as Nb_2O_5 .

Yet another embodiment of the present invention is a power supply or a
15 core for a transformer that includes the ferrite material according to the compositions described immediately above.

Another embodiment of the present invention is a sintered material that may include the ferrite material according to the compositions described immediately above, having a Curie temperature greater than 250°C ,

20 Yet another embodiment of the present invention is a sintered material that includes the ferrite material and has a power loss of below 170 mW/cm^3 at a frequency of 0.5 MHz and a magnetic flux density of 500 G, a power loss of below 465 mW/cm^3 at a frequency of 1.0 MHz and a magnetic flux density of 500 G, or a power loss of below 300 mW/cm^3 at a frequency of 3.0 MHz and a
25 magnetic flux density of 100 G, at a temperature range between 80°C and 140°C .

Another embodiment of the present invention is a sintered material having a power loss at or below 100 mW/cm^3 and a temperature between 80°C and 140°C and a frequency of 250 kHz.

Another embodiment of the present invention provides a ferrite material consisting essentially of, as main components, an amount of iron oxide ranging from 55.5 to 58 mole percent calculated as Fe_2O_3 , an amount of manganese oxide ranging from 38.1 to 40.5 mole percent calculated as MnO , and an amount of zinc oxide ranging from 3.3 to 4.7 mole percent calculated as ZnO , and, as minor components, an amount of calcium oxide ranging from 0.035 to 0.100 mole percent calculated as CaO , an amount of silicon oxide ranging from 0.020 to 0.040 mole percent calculated as SiO_2 , and an amount of niobium oxide ranging from 0.010 to 0.030 mole percent calculated as Nb_2O_5 .

10 In addition, the present invention is directed to a method of forming a ferrite material that includes mixing as main components, an iron component ranging from 55.5 to 58.0 mole percent calculated as Fe_2O_3 , an amount of manganese component ranging from 38.0 to 41.0 mole percent calculated as MnO , and an amount of zinc component ranging from 3.3 to 4.7 mole percent calculated as ZnO , and mixing as minor components an amount of calcium component ranging from 0.030 to 0.100 weight percent calculated as CaO , an amount of silicon component ranging from 0.015 to 0.040 weight percent calculated as SiO_2 , and an amount of niobium component ranging from 0.010 to 0.030 weight percent calculated as Nb_2O_5 .

15 Yet another embodiment of the present invention is a method of forming a core material that includes mixing as main components, an iron component ranging from 55.5 to 58.0 mole percent calculated as Fe_2O_3 , an amount of manganese component ranging from 38.0 to 41.0 mole percent calculated as MnO , and an amount of zinc component ranging from 3.3 to 4.7 mole percent calculated as ZnO , and mixing with the main components minor components comprising an amount of calcium component ranging from 0.030 to 0.100 weight percent calculated as CaO , an amount of silicon component ranging from 0.015 to 0.040 weight percent calculated as SiO_2 , and an amount of niobium component ranging from 0.010 to 0.030 weight percent calculated as

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Nb₂O₅, the main components and the minor components forming a ferrite material. The ferrite material may be pressed to a predetermined density, and sintered to form the core material.

It should be understood that this invention is not limited to the
5 embodiments disclosed in this Summary, but it is intended to cover
modifications that are within the spirit and scope of the invention, as defined by
the appended claims.

10 **Detailed Description of the Preferred Embodiments**

It is to be understood that certain descriptions of the present invention
have been simplified to illustrate only those elements and limitations that are
relevant to a clear understanding of the present invention, while eliminating, for
purposes of clarity, other elements. Those of ordinary skill in the art, upon
15 considering the present description of the invention, will recognize that other
elements and/or limitations may be desirable in order to implement the present
invention. However, because such other elements and/or limitations may be
readily ascertained by one of ordinary skill upon considering the present
description of the invention, and are not necessary for a complete
20 understanding of the present invention, a discussion of such elements and
limitations is not provided herein. For example, as discussed herein, the
materials of the present invention may be incorporated, for example, as core
materials for coils or transformers in various power supplies, and the like. Core
materials for coils or transformers are understood by those of ordinary skill in
25 the art, and, accordingly, are not described in detail herein.

Furthermore, compositions of the present invention will be generally
described in the form of a manganese-zinc ferrite material that may be
incorporated as high frequency core materials. It will be understood, however,
that the present invention may be embodied in forms and applied to end uses

that are not specifically and expressly described herein. For example, one skilled in the art will appreciate that the present invention may be incorporated into high frequency devices other than core materials that are not specifically identified herein.

5 The term "pulverization," as used herein, refers to mechanically dividing, fragmenting, or disintegrating a material (such as zinc, manganese, and iron oxides or compounds thereof) or other material into a powder. In the method of the present invention, pulverization may be carried out in a manner that provides the resultant powder particles with a desired particle size as described
10 hereinbelow. As used herein, pulverization includes, for example, all forms of mechanically dividing, fragmenting, or disintegrating a larger mass into a powder, including atomization, crushing, milling, grinding, cold stream processing, and the like.

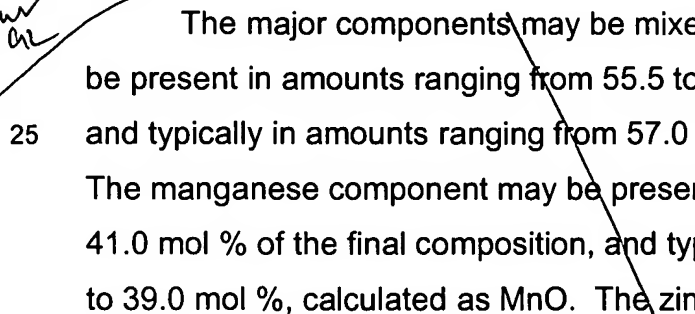
 Other than in the operating examples, or unless otherwise expressly
15 specified, all of the numerical ranges, amounts, values and percentages such as those for amounts of materials, times and temperatures of reaction, ratios of amounts, and others in the following portion of the specification may be read as if prefaced by the word "about" even though the term "about" may not expressly appear with the value, amount, or range. Accordingly, unless indicated to the
20 contrary, the numerical parameters set forth in the following specification and attached claims are approximations that may vary depending upon the desired properties sought to be obtained by the present invention. At the very least, and not as an attempt to limit the application of the doctrine of equivalents to the scope of the claims, each numerical parameter should at least be construed
25 in light of the number of reported significant digits and by applying ordinary rounding techniques.

 Notwithstanding that the numerical ranges and parameters setting forth the broad scope of the invention are approximations, the numerical values set forth in the specific examples are reported as precisely as possible. Any

numerical value, however, inherently contains certain errors necessarily resulting from the standard deviation found in their respective testing measurements. Furthermore, when numerical ranges of varying scope are set forth herein, it is contemplated that any combination of these values inclusive of
5 the recited values may be used.

The present invention is directed, generally, to ferrite materials, and more particularly, to manganese-zinc ("Mn-Zn") magnetic ferrite materials and methods of forming and employing the same that are designed to operate over a wide range of high frequencies and/or high temperatures with low power
10 losses. It has been found that low power losses at high frequency and high temperature may be obtained from Mn-Zn materials that are a combination of major components and minor components at particular sintering conditions, as set forth herein. The materials of the present invention include oxides or compounds that convert into oxides upon heating. These materials include
15 components of iron, manganese, and zinc, as major components, and components of silicon, calcium, and niobium (i.e. columbium), as minor components.

As illustrated in the working examples set forth herein, typically the Mn-Zn ferrite compositions of the present invention contain oxides of iron, such as
20 Fe_2O_3 , manganese, such as MnO , and zinc, such as ZnO , as major components, and combine amounts of oxides of calcium, such as CaO , silicon, such as SiO_2 , and niobium, such as Nb_2O_5 , as minor components.

The major components may be mixed such that the iron component may be present in amounts ranging from 55.5 to 58.0 mol % of the final composition, and typically in amounts ranging from 57.0 to 57.3 mol %, calculated as Fe_2O_3 .
25 The manganese component may be present in amounts ranging from 38.0 to 41.0 mol % of the final composition, and typically in amounts ranging from 37.0 to 39.0 mol %, calculated as MnO . The zinc component may be present in

amounts ranging from 3.3 to 4.7 mol % of the final composition, and typically in amounts ranging from 4.0 to 4.7 mol %, calculated as ZnO.

To the major components are added trace, but effective, amounts of oxides (or carbonates) of calcium, silicon, and niobium. The calcium
5 component may be added to the major components in amounts ranging from 0.030 to 0.100 wt %, based on the total weight of the ferrite material, and typically in amounts ranging from 0.030 to 0.050 wt %, calculated as CaO. The silicon component may be added to the major components in amounts ranging from 0.015 to 0.040 wt %, based on the total weight of the ferrite material, and
10 typically in amounts ranging from 0.015 to 0.035 wt %, calculated as SiO₂. The niobium component may be added to the major components in amounts ranging from 0.010 to 0.030 wt %, and typically in amounts ranging from 0.020 to 0.030 wt %, based on the total weight of the ferrite material, calculated as Nb₂O₅.
15 It will be appreciated by one of ordinary skill in the art that although specific oxides for each metal component are discussed herein, other suitable oxides (or carbonates) of iron, manganese, zinc, silicon, calcium, and niobium, if applicable, may be used to form the ferrite materials of the present invention. Accordingly, although the particular metal oxides disclosed herein (ZnO, MnO, Fe₂O₃, CaO, SiO₂, and Nb₂O₅) have been found to provide good results in
20 embodiments of the present invention, one of skill in the art would understand that the present invention need not be limited to the use of the specific oxidation state identified, and that other metal oxides of other oxidation states or their carbonates may be employed as a partial or complete substitute for the particular metal oxide. For example, with respect to iron oxide, the present
25 invention may employ FeO, Fe₂O₃, and Fe₃O₄, and compounds capable of being converted into Fe₂O₃, such as iron hydroxide, iron oxalate, and the like; with respect to manganese oxide, the present invention may employ MnO, MnO₂, Mn₃O₄, and compounds capable of being converted into MnO, such as

manganese carbonate, manganese oxalate, and the like; with respect to zinc oxide, the present invention may employ ZnO, and compounds capable of converting into ZnO, such as zinc carbonate, zinc oxalate, and the like.

Accordingly, although specific metal oxides are reported to describe the components of the present invention, one of ordinary skill in the art will understand that the scope of the present invention need not be limited to only these specific components.

The ferrite materials of the present invention and the products that incorporate the same may be formed by mixing oxides or carbonates of iron, manganese, and zinc as starting materials in the amounts discussed above. Raw materials of iron, manganese, and zinc oxides or carbonates may be mixed before or after pulverization in any manner known to those of ordinary skill in the art, such as through dry blending. The raw materials may be pulverized, such as through grinding, to a particle size ranging from 0.9 μm to 1.9 μm . The raw materials often show variations in the contents of the desired components, which must be monitored and adjusted, if necessary, to the appropriate mole or weight percentage discussed above, because the sintering behavior and resultant material properties are affected by the amounts of these components.

A dispersant, such as Lomar®, commercially available from Henkel Corporation, Morristown, N.J., may be added to the dry blend along with water to form a slurry. Other dispersants known to those skilled in the art may be employed as long as the dispersant employed is relatively pure and the amount of trace impurities that may be added to the ferrite system is limited. When a dispersant is employed, the specific dispersant to raw component ratio can vary widely so long as it provides the requisite or desired viscosity for grinding, with amounts typically ranging from 0.8 wt % to 1.2 wt %.

Oxides or carbonates of calcium, silicon, and niobium may be added to the slurry in the amounts discussed above. Because relatively small amounts

of each of the minor components is employed, these components are typically added in pure (i.e. at least 99.9%) powder oxide form, rather than in the raw bulk form that is used to form the blend of major components.

Additives such as polyvinyl alcohol and glycerin may be added to the
5 slurry composition prior to milling that act as sacrificial binder materials for the pressed form. Other binder materials known to those skilled in the art may be employed as long as the binding agent chosen satisfies the relatively strict purity standards that limit the amount of trace impurities that are added to the ferrite system. Although the amount of binder material that may be added to
10 the system can vary widely, when polyvinyl alcohol and glycerin are employed, both binder materials are typically added in amounts ranging from 1.0 to 2.0 wt %.

The slurry may be milled and spray dried to produce a granulated powder for pressing into core shapes having a predetermined shape, size, and
15 pressed density. Typical pressed shapes include, for example, toroids, planar E-cores, and pot cores. The density of the pressed shapes typically ranges from 3.1 to 3.3 g/cm³. The cores may be sintered at temperatures ranging from 1130°C to 1180°C and then cooled to temperatures ranging from 20°C to 30°C to form the sintered core material. The oxygen content of the atmosphere
20 should be controlled during the cooling portion of the cycle based on the temperature and rate of cooling, as known to those of ordinary skill in the art.

As illustrated in the Example and Tables 1 and 2, it has been found that the Mn-Zn ferrite materials that combine components of iron, such as Fe₂O₃, manganese, such as MnO, and zinc, such as ZnO, as major components, and
25 components of silicon, such as SiO₂, calcium, such as CaO, and niobium, such as Nb₂O₅, as minor components, in the amounts discussed above, provide improved properties relative to known ferrite materials. In particular, the compositions of the present invention have relatively low ZnO content and a high Fe₂O₃ content compared to typical Mn-Zn ferrites, and further combines

amounts of CaO, SiO₂, and Nb₂O₅ to control and limit power losses. It has been found that the combination of these components result in a material having improved ferrite properties.

5 The present invention will be described further by reference to the following example. The following examples are merely illustrative of the invention and are not intended to be limiting. Unless otherwise indicated, all parts are by weight.

Examples

10 Raw materials of Fe, Mn, and Zn oxides were dry blended and ground 1 1/2 hours in an attrition mill to an average particle size of approximately 1.40 μm. An addition of 1.0 weight percent Lomar® (Henkel Corporation) was added at the beginning of the grinding operation to act as a dispersant. Oxides or carbonates of Ca, Si, and Nb were also added at the beginning of the grinding
15 operation. Before the resulting slurry was removed from the mill, 1.5 weight percent polyvinyl alcohol and 1.5 weight percent glycerin were added to the slurry. The slurry was milled for another 15 minutes and then spray dried to produce a granulated powder for pressing.

20 Test cores were pressed in the shape of toroids having an outside diameter of 22mm, an inside diameter of 13.7 mm and a height of 6.3 mm.

The pressed density of the cores was 3.20 g/cc. The test cores were sintered at 1160°C for 5 hours. The oxygen content of the atmosphere was controlled during the cooling portion of the cycle. Temperatures as low as 1130°C have been found to form acceptable material.

25 Compositions of the present invention are listed below in Table 1.

TABLE 1

Lot No.	Fe ₂ O ₃ (mol%)	MnO (mol%)	ZnO (mol%)	CaO (wt%)	SiO ₂ (wt%)	Nb ₂ O ₅ (wt%)	Co ₃ O ₄ (wt%)	TiO ₂ (wt%)	Other (wt%)
1	56.90	39.05	4.05	0.045	0.020	0.030	---	---	---
2	56.90	39.05	4.05	0.090	0.020	0.030	---	---	---
3	57.00	39.00	4.00	0.040	0.020	0.025	---	---	---
4	57.50	38.55	3.95	0.037	0.020	0.027	---	---	---
5	56.85	39.20	3.95	0.037	0.020	0.027	---	---	---
6	57.45	38.15	4.40	0.037	0.020	0.027	---	---	---
7	56.40	39.15	4.45	0.037	0.020	0.027	---	---	---
8	55.55	40.50	3.95	0.037	0.020	0.027	---	---	---
9	56.80	39.85	3.35	0.037	0.020	0.027	---	---	---
10	56.35	40.15	3.50	0.037	0.020	0.027	---	---	---
11	56.80	39.15	4.05	0.037	0.020	0.027	---	---	---
12	57.45	39.00	3.55	0.037	0.020	0.027	---	---	---
13	56.95	38.35	4.70	0.037	0.020	0.027	---	---	---
14	57.05	38.75	4.20	0.040	0.025	0.025	---	---	---
15	57.15	38.15	4.70	0.040	0.025	0.025	---	---	---
16	57.15	38.15	4.70	0.040	0.020	0.040	---	---	---
17	57.15	38.15	4.70	0.040	0.020	0.010	---	---	---
18	57.15	38.15	4.70	0.040	0.030	0.025	---	---	---
19	57.15	38.15	4.70	0.040	0.010	0.025	---	---	---
Comparative Examples									
20	57.00	39.00	4.00	0.100	0.020	---	---	---	0.05 Ta ₂ O ₅
21	57.00	39.00	4.00	0.100	0.020	---	---	---	0.05 GeO ₂
22	55.00	39.00	6.00	0.100	0.020	---	---	---	0.05 GeO ₂
23	57.45	36.98	5.57	0.098	0.030	0.020	0.184	0.331	---
24	56.48	38.00	5.52	0.105	0.027	0.022	0.195	0.351	---
25	56.95	38.85	4.20	0.040	0.060	0.030	---	---	---
26	53.50	33.99	11.38	0.056	0.020	0.016	0.149	0.580	---

TABLE 2

Lot No.	Perm @ 100kHz	Bmax @ 15 Oe. 25°C (kG)	Bmax @ 15 Oe. 100°C (kG)	Temp. of Min Watt Loss (°C) ¹	Watt Loss @ 0.5 MHz, 500G (mw/cc) ¹	Watt Loss @ 1 MHz, 500G (mw/cc) ¹	Watt Loss @ 3 MHz, 100G (mw/cc) ¹	Curie Temp (°C)
1	830	5.00	4.45	120	100	350	130	280
2	790	5.05	4.45	120	170	465	160	280
3	745	5.05	4.41	120	105	375	130	285
4	735	5.20	4.45	100	110	400	105	290
5	725	5.10	4.40	120	110	400	125	285
6	755	5.15	4.45	100	115	380	105	285
7	740	5.05	4.35	120	118	420	165	280
8	630	4.95	4.30	140	160	630	300	270
9	670	5.10	4.45	120	130	430	135	290
10	660	5.05	4.35	140	120	450	180	270
11	735	5.15	4.40	120	105	390	125	285
12	710	5.15	4.55	100	115	395	105	285
13	780	5.15	4.45	120	105	375	120	280
14	750	5.20	4.55	120	95	320	100	280
15	800	5.25	4.55	100	85	315	90	280
16	570	5.30	4.61	110	533	1625	725	280
17	825	5.15	4.46	100	80	310	120	280
18	380	5.35	4.68	100	1125	4650	1390	280
19	800	5.07	4.39	110	85	330	125	280
Comparative Examples								
20	760	5.00	4.37	120	100	440	220	275
21	785	5.21	4.51	100	113	500	205	275
22	734	4.59	3.86	100	267	1005	310	250
23	224	3.54	3.05	40	2190	5492	200	250
24	980	5.21	4.34	25	110	360	35	270
25	475	5.30	4.64	120	480	1340	360	270
26	1470	4.60	3.90	60	109	494	244	240

¹ All watt loss measurements were taken at the temperature where the minimum watt loss occurs.

As shown in the Example, in corresponding Tables 1 and 2, and as discussed above, the ferrite materials of the present invention combine components of iron, such as Fe_2O_3 , manganese, such as MnO , and zinc, such as ZnO , as major components, and components of silicon, such as SiO_2 , calcium, such as CaO , and niobium, such as Nb_2O_5 , as minor components, in specified amounts, to provide improved properties relative to known ferrite materials. The present invention demonstrates that large additions of CaO and other components ultimately serve to increase the power losses at high frequencies. Lot No. 25, for example, is provided as a comparative example, and illustrates the poor results obtained when a composition having component amounts (in this case 0.06 wt % SiO_2) outside of the disclosed ranges are prepared and tested.

Compositions of the present invention control and limit power losses and result in a material having a low permeability and a high Curie temperature. The materials of the present invention can limit, or substantially eliminate, the addition of large amounts of Co_3O_4 , SnO_2 , TiO_2 , CaO , and the like, or combinations of these components, while achieving exceptionally good material properties. Accordingly, sintering the combination of major and minor components provides ferrite materials of the present invention having more consistent material properties relative to known compositions because of their lesser degree of sensitivity to firing conditions.

It has been found that embodiments of the present invention provide a ferrite material with a permeability (μ) of less than 830, and more typically ranging from 700 to 800 at 25°C . In addition, it has been found that embodiments of the present invention provide a ferrite material with a Curie temperature of greater than 250°C , typically 270°C or greater, and more particularly 280°C or greater.

The relatively high Curie temperatures obtained in embodiments of the present invention also improve the flux density versus temperature response of

the material. Measurements taken of embodiments of the present invention indicate maximum magnetic flux densities (B_{\max}) range from 3500 to 4500 G, and are typically greater than 4000 G, at 100°C, which is significantly higher than known Mn-Zn ferrite materials designed for high frequency applications of
5 between 250 kHz to 3MHz.

As illustrated in the test results, at temperatures between 100°C and 140°C, embodiments of the present invention limit, or substantially reduce, the power loss at 0.5 MHz, 1 MHz, and 3MHz, relative to known ferrite materials. Measurements of power losses of embodiments of the present invention
10 indicate that power losses in certain embodiments are below 100 mW/cm³.

Power losses of sintered compositions of the present invention were measured by winding the toroid test samples with the appropriate number of turns and then applying a sine wave voltage at the desired frequency and at an amplitude sufficient to generate the desired flux density in the core. The current
15 (I) required to achieve the set voltage (V) was then measured as was the phase angle (θ) between the applied voltage and the measured current. Power losses are expressed as: $P = VI \cos \theta$. The power in watts is divided by the volume of the test specimen to obtain a normalized power loss in milliwatts per cubic centimeter of material (mW/cm³). This loss measurement includes losses due
20 to the copper windings, which were assumed to be small.

Power losses at 0.5 MHz were measured to be below 170 mW/cm³, and typically ranged from 85 mW/cm³ to 130 mW/cm³. In particular, test results show that the ferrite compositions of the present invention may be used to form a sintered material comprised of the ferrite material having a power loss of
25 below 170 mW/cm³ at a frequency of 0.5 MHz and a magnetic flux density of 500 G at a temperature range between 80°C and 140°C

Power losses at 1.0 MHz were typically measured to be below 465 mW/cm³, and more particularly ranged from 315 mW/cm³ to 400 mW/cm³. In particular, test results show that the ferrite compositions of the present invention

may be used to form a sintered material comprised of the ferrite material having a power loss of below 465 mW/cm^3 at a frequency of 1.0 MHz and a magnetic flux density of 500 G at a temperature range between 80°C and 140°C .

Power losses at 3.0 MHz were measured to be below 300 mW/cm^3 , and
5 more typically ranged from 90 mW/cm^3 to 180 mW/cm^3 . In particular, test results show that the ferrite compositions of the present invention may be used to form a sintered material comprised of the ferrite material and having a power loss of below 300 mW/cm^3 at a frequency of 3.0 MHz and a magnetic flux density of 100 G at a temperature range between 80°C and 140°C .

10 In some embodiments, test results show that the ferrite materials of the present invention may be used to form a sintered material having a power loss at or below 100 mW/cm^3 at a temperature between 80°C and 140°C and a frequency of 250 kHz.

It has been found that certain embodiments of the present invention have
15 been successful at simultaneously achieving low power losses above 500 kHz, minimum power loss at a temperature of 100°C or greater, saturation magnetic flux density greater than 4300 G at 100°C , a Curie temperature greater than 270°C , while providing relative ease of processing. Prior art techniques have not been successful in meeting this combination of the performance criteria.

20 The present invention not only meets all of these requirements but also uses a much simpler composition than anything found in the prior art. The end result is a magnetic material with superior performance that is simple to process.

Compositions of the present invention that use the combination of the oxides (or carbonates) described herein, particularly those compositions that
25 employ Nb_2O_5 and relatively low amounts of CaO , result in ferrite materials having improved chemical and physical properties, as illustrated in Example 1 and Tables 1 and 2. Accordingly, improvements to the magnetic flux density, and power loss at high operating frequencies between 250 kHz to 3MHz allow

reduction in the magnetic cross section of the core materials that employ the compositions of the present invention.

Test results of the materials of the present invention show that there is another composition region in which B_{\max} can be increased above 5000 G at 25°C and increased above 4000 G at 100°C. It has also been found that relative to the materials of the present invention, prior art materials do not achieve the same level of performance and are more difficult to process. The present invention demonstrates that fewer additive additions at much lower concentrations are more effective at reducing power losses even though the resistivity of the material may not be increased. Also, as shown in the comparative results below, a reduction in grain size is not a prerequisite for low losses.

Lot No.	Grain Size	Resistivity @ 1 MHz	Watt Loss @ 1 MHz, 500G
15	8-9 μm	118 ohm-cm	315 (mW/cm ³)
Comparative Example 26	3-4 μm	248 ohm-cm	494 (mW/cm ³)

Although embodiments of the present invention identified in the Example are identified as only having components of iron, such as Fe_2O_3 , manganese, such as MnO , and zinc, such as ZnO , as major components, and components of silicon, such as SiO_2 , calcium, such as CaO , and niobium, such as Nb_2O_5 , as minor components, it is contemplated that the Mn-Zn ferrite materials of the present invention may include these components, ranging in amounts described herein, as well as other minor components known to those skilled in the art that impart desirable properties to the material.

It will be appreciated by those of ordinary skill in the art that the improved properties of the materials of the present invention allow these materials to be incorporated into products that require high frequency operation, such as those

related to commercial switching power supplies, as well as for household electric appliances, communication and telecommunication equipment, computer and peripheral equipment, electronics finished products, electronic components, down hole oil drilling sensors, automotive applications, and other high frequency electronic circuitry. Embodiments of the present invention provide for high temperature operation of 100°C or higher, with low power losses at high flux densities and high frequencies that provide increased power performance to be added to products having a limited area, that will allow for miniaturization of the core volume.

It will also be appreciated by those skilled in the art that changes could be made to the embodiments described above without departing from the broad inventive concept thereof. It is understood, therefore, that this invention is not limited to the particular embodiments disclosed, but it is intended to cover modifications that are within the spirit and scope of the invention, as defined by the appended claims.